AUTHORS:

Andreyeva, N. S., Iveronova, V. I.,

62-58-3-27/30

Kozarenko, T. D., Poroshin, K. T., Shibnev, V. A., Shutskever, N. Ye.

TITLE:

Investigation of the Structure of Peptides Containing

Glycine and 1-Proline (Issledovaniye struktury peptidov,

soderzhashchikh glitsin i 1-prolin)

PERIODICAL:

Izvestiya Akademii Nauk SSSR,Otdeleniye Khimicheskikh

Nauk, 1958, Nr 3, pp. 376-377 (USSR)

ABSTRACT:

The investigation of peptides containing amino acids is of importance for the investigations of the structure of proteins. The stereochemical rôle of pyrrolidine rings within the configuration of the polypeptide chain has not yet been sufficiently explained. In general it is assumed that the bends of the polypeptide chains are formed in such points, where residues of proline and oxyproline are present. At present structural investigations of the peptides and polypeptides of numerous amino acids are carried out. There have, however, only few works been published on the investigation of compounds containing amino acids. The

Card 1/2

aim of this work is the investigation of the above mentioned

62-58-3-27/30 Investigation of the Structure of Peptides Containing Glycine and 1-Proline

structure of peptides. Glycyl-1-prolyl, 1-prolylglycine, carbobenzoxyglycyl-1-prolyl and the anhydride of glycyl-1proline were synthetized. Furthermore the first stage of the x-ray analysis of the synthetized compounds was

There are 1 table and 10 references, 1 of which is Soviet.

ASSOCIATION:

Fizicheskiy fakul tet Moskovskogo gosudarstvennogo universiteta i Institut organicheskoy khimii im.

N. D. Zelinskogo Akademii nauk SSSE

Physics Department of Moscow State University and the Institute for Organic Chemistry imeni N. D. Zelinskiy,

AS USSR)

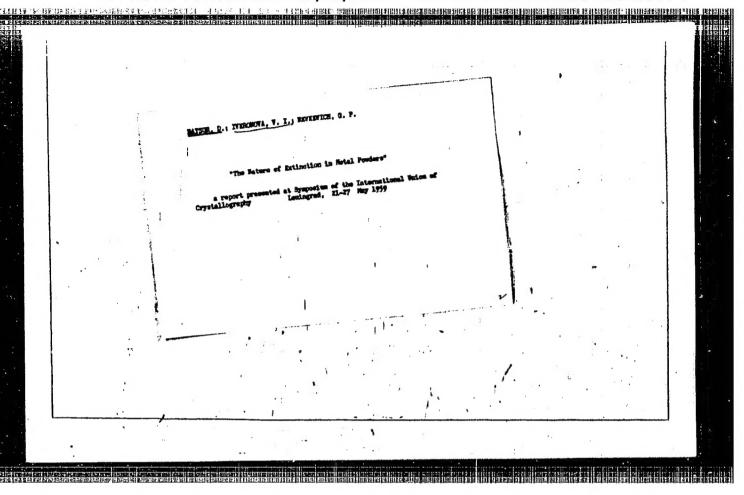
SUBMITTED:

October 31, 1957

Card 2/2

APPROVED FOR RELEASE: 03/20/2001 CIA-RDP86-00513R000619320004-6"

"APPROVED FOR RELEASE: 03/20/2001 CIA-RDP86-00513R000619320004-6



SOV/70-4-1-4/26 Iveronova, V. I. and Katsnel'son, A. A. AUTHORS:

Short-range Order and Characteristic Temperature of TITLE:

the Alloy NigPt (Blizhniy poryadok i kharakteristicheskaya

ः राज्य तम्बन्धानम् अवविभावकन्तराक्षारम् व विवासम्बन्धानस्य विभावतस्य । विद्यासस्य विभावतस्य विभावतस्य । विद्य

temperatura splava NizPt)

PERIODICAL: Kristallografiya, 1959, Vol 4, Nr 1, pp 25-29 (USSR)

Parallel measurements of short-range order and character-ABSTRACT: istic temperature of the alloy NizPt have been made.

It has been established that annealing for four hours at 700°C after plastic deformation leads to an increase in characteristic temperature and short-range order, It is proposed that changes in the characteristic temperature should not be connected only with changes in short-range order. A hypothesis on the influence

of "the Cotterell atmosphere" in changing the

characteristic temperature in certain alloys is enunciated.

The polycrystalline NizPt specimens were prepared by melting in an h.f. furnace under argon. They were homogenised for 50 hours at 900°C. The concentration of

25.5 at.% Pt was found from the lattice dimensions.

Plastic deformation was produced by coarse polishing Card1/3

SOV/70-4-1-4/26

Short-range Order and Characteristic Temperature of the Alloy

NizPt

and was followed by 1 and 20 hours at 700°C in a vacuum furnace. The diffuse scattering was measured with a URS-50I diffractometer with FeK, radiation monochromatised in the reflected beam. The diffuse scattering is due to (a) scattering connected with correlation in the positions of atoms in the alloy (short-range order, segregation, etc.), (b) scattering due to atomic displacements because of differences in the atomic radii of the components, and (c) scattering from the overlapping of the tails of the lines due to lattice distortion on plastic deformation. These were separated because of their different angular dependences. Characteristic temperature was measured from powder photographs (Zr-filtered Mo radiation) taken at +20 to -160°C. For various treatments the results were:

Plastically deformed  $\alpha_1 = -0.06\pm0.02$ ,  $n_1^{\text{PtNi}} = 9.5$ ,  $\theta = 315^{\circ} \pm 15^{\circ}$ 450°+50° 360°+20° 10.7, Annealed 700° 1 hour 700° 20 hours  $-0.19\pm0.01$ ,  $-0.18\pm0.01$ , 10.7, 12.0 -0.33 Fully ordered

Card2/3

Fully disordered

Short-range Order and Characteristic Temperature of the Alloy NigPt

Here  $\alpha_1$  is the short-range order parameters for the first sphere,  $n_1^{\rm PtN1}$  is number of N1 atoms in the first sphere round a Pt atom. In the plastically deformed state 3 agrees within the experimental limits with

 $\left[\frac{3}{4}\,\,\mathrm{m_{Ni}}\,+\frac{1}{4}\,\,\mathrm{m_{Pt}}\right]\Theta^{2}=\frac{3}{4}\,\,\mathrm{m_{Ni}}\Theta^{2}_{\mathrm{Ni}}\,+\frac{1}{4}\,\,\mathrm{m_{Pt}}\Theta^{2}_{\mathrm{Pt}},$ 

where  $m_{\mbox{Ni}}$  and  $m_{\mbox{Pt}}$  are the atomic concentrations. There are 1 figure, 1 table and 21 references, 13 of which are Soviet, 4 English and 4 international.

ASSOCIATION: Moskovskiy gosudarstvennyy universitet imeni M. V. Lomonosova (Moscow State University imeni M. V. Lomonosov)

SUBMITTED: July 10, 1958

Card3/3

SOV/70-4-2-12/36

AUTHORS: Batsur', D., Iveronova, V.I. and Revkevich, G.P.

TITLE: The Nature of Extinction in Metallic Fowders (Priroda

ekstinktsii v metallicheskikh poroshkakh)

PERIODICAL: Kristallografiya, 1959, Vol 4, Nr 2, pp 214-218 (USSR)

ABSTRACT: X-ray scattering curves from powdered Cu and Ni have been measured with an URS-50I diffractometer for

Cu-radiation monochromatised by reflexion from pentaerithritol. These are compared with theoretical curves. It is concluded that in powders of Cu and Ni deformed and annealed below the recrystallisation temperature secondary extinction is observed. The coefficient of secondary extinction grows with increasing temperature of annealing which corresponds to decreasing the discrientation angle. The dimensions of the blocks here increases very little and primary extinction can be neglected. After high-temperature annealing (above the recrystallisation temperature which leads to a sharp growth of the grains) only primary extinction influences the intensities on the powder photograph and secondary extinction is negligible. The primary extinction is

Card1/3

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SOV/70-4-2-12/36

The Nature of Extinction in Metallic Powders

readily detectable from the weakening of all lines, even. those with high indices. This shows that in powder specimens blocks in the same grain screen each other and not blocks in different grains. The disorientation of blocks in one grain of the deformed metal is comparatively slight (a general fragmentation of 1...5°) but the number of blocks is large and hence there is secondary extinction. After recrystallisation the number of blocks in a grain has become small but the angles of rotation between them are large and this removes secondary extinction and produces primary. To determine the dimensions of the blocks from the intensities of the Debye lines it is necessary first to make certain what sort of extinction has weakened the intensities. Coincidence between the dimensions determined by the two methods (line broadening and intensity measurement) which are noted in the literature are only apparent and are caused by the irregular use of the formulae for the dependence of intensity on block size for specimens where the weakening of lines is due to the slight fragmentation of the initial grains.

Card2/3

APPROVED FOR RELEASE: 03/20/2001 CIA-RDP86-00513R000619320004-6"

SOV/70-4-2-12/36 The Nature of Extinction in Metallic Powders

There are 5 figures and 10 references, 4 of which are Soviet and 6 English

ASSOCIATION: Moskovskiy gosudarstvennyy universitet imeni

M.V. Lomonosova (Moscow State University imeni

M.V. Lomonosov)

SUBMITTED: September 13, 1958

Card 3/3

### "APPROVED FOR RELEASE: 03/20/2001 CIA-RDP86-00513R000619320004-6 TO THE POST OF BUILDING AND A STREET OF THE PARTY OF THE

SOV/48-23-5-12/31

24(3) AUTHORS: Batsur', D., Iveronova, V. I., Revkevich, G. P.

TITLE:

On the Problem of Tensions of the 3rd Kind (K voprosu o

napryazheniyakh III roda)

PERIODICAL:

Izvestiya Akademii nauk SSSR, Seriya fizicheskaya, 1959,

Vol 23, Nr 5, pp 591-600 (USSR)

ABSTRACT:

N. N. Davidenkov has shown that strains (or tensions) of the 3rd arise by plastic deformation. In a series of non-Russian papers it is proven that in roentgenograms of plastically deformed metals an attenuation of the interference lines occurs, and a formula (1) is given, permitting the computation of this attenuation. Reference is then made to works carried out in the Forties, when it was proven that tensions of the 3rd kind occur with all deformations. An explanation is given next of two models of the state of plastically deformed metals 1) the dislocation is in the range of the grain boundaries of the texture. 2) the disordered dislocation is in the interior of the texture grains. The attenuation of the interference lines is in relation to the root mean square atom displacement. Mention is then made of conditions in experiments, in which the monochromatic radiation was applied and measurements for comparative purposes were made

Card 1/2

On the Problem of Tensions of the 3rd Kind

SOV/48-23-5-12/31

on standard samples. The results of intensity measurements at a temperature of 500°C are first given next. A strong increase in intensity is observed at the beginning of the thermal treatment. A formula is then given, by which the degree of deformation may be determined from the intensity of the lines. Ekstein's formula is given for the computation of the primary extinction (Ref 12), and Lang's formula for the secondary extinction (Ref 13). Measuring results are shown in diagrams, that were obtained from experiments made on nickel, copper and the alloy Cu-Sn (Figs 3 and 4), and the dispersion was investigated. The conclusive summary mentions the relations between mechanical characteristics and tensions of the 3rd kind. There are 4 figures, 2 tables, and 19 references, 11 of which are Soviet.

Card 2/2

IVERONOVA, V. I. and KATSNELSON, A. A.

"Short Order and X-ray Debye Temperature of Ni<sub>3</sub>Pt."

report presented at the Fifth International Congress of the International Union of Crystallography, Cambridge, UK, 15-24 Aug 1960

18.8100

77599 SOV/148-60-1-22/34

AUTHORS:

Iveronova, V. I., Kassandrova, O. N., Rozantseva, Ye. G.

TTTLE:

Characteristic Temperature of Iron-Vanadium Alloys

PERIODICAL:

Izvestiya vysshikh uchebnykh zavedeniy. Chernaya

metallurgiya, 1960, Nº 1, pp 133-135 (USSR)

ABSTRACT:

The accuracy of the expression of interatomic bond strength in terms of a characteristic temperature, computed according to the difference between the X-ray

diffraction intensities at different temperatures, has

been questioned by a number of investigators

(Kristallografiya, Vol 22, 444, 1957; Izv. AN SSSR, Section of Physics, 20, 723, 1956; etc.), The reason for this was the appreciable dependence of diffraction intensities on preceeding treatments. Similarly, proportionality of modulus of elasticity to characteristic

temperature during heat treatment was found to be qualitative rather than quantitative (Fizika metallovi metallovedeniye, Vol 4, 417, 1957). It became known

Card 1/3

Characteristic Temperature of Iron-Vanadium Alloys

77599 SOV/148-50-1-22**/**34

that the exponent in a temperature-factor expression actually depends not only on the maximum frequency of oscillations, related to moduli of elasticity produt also on the spectral distribtuion of thermal vibrations which is not considered to be a variable in the Debye-Waller equation. Thus, the authors J determined the characteristic temperature of ironvanadium alloys by the X-ray method and compared the found values with those determined earlier from the moduli of elasticity, in order to evaluate the extent to which the thermal-vibration spectrum changes with the change in the alloy composition. Similar earlier attempts produced contradictory results. The iron-base alloys with 6 to 28% V were machined into 1 mm cylinders, annealed under vacuum at 600° C for 1 hr, cooled off, and 7 X-ray diffraction photographs were taken from each cylinder at -150° C to 220° C. The computed characteristic temperatures, 1.e., 475, 485 and 540° C for the a-phase containing 6.1, 1/2. 17.2, and 28.8% V, respectively, proved to increase

Card 2/3

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Characteristic Temperature of Tron-Vanadium Alloys

77699 SOV/148-60-1-22/34

with the V content. Above 28.8% V the \$\mathcal{O}\$-phase appears, and its weak diffraction lines could already be seen on overexposed photographs. It seems that the frequency spectrum of iron-vanadium alloys is close to that of one-atom structures because of only a slight difference between the atomic weights of Fe and V. At the same time, the interatomic bond strength in the structure of iron-vanadium alloys is higher than that of pure iron. This conforms with the fact, established by N. S. Rysina and B. N. Finkel shteyn, that the shear modulus of iron-vanadium alloys drops with temperature increase at a slower rate than of pure iron. There is I figure; I table; and 6 Soviet references.

ASSOCIATION:

Moscow State University (Moskovskiy gosudarstvenyy

universitet)

SUBMITTED:

September 16, 1958

Card 3/3

S/139/60/000/006/015/032 E032/E414

Iveronova, V.I. and Zvyagina, A.P. AUTHORS:

Determination of the Characteristic (Debye)

Temperature by X-Ray Methods TITLE:

PERIODICAL: Izvestiya vysshikh uchebnykh zavedeniy, Fizika,

1960, No.6, pp.105-108

The X-ray method of determination of the Debye temperature of alloys is based on an application of the theory of X-ray scattering due to Debye. This theory incorporates the following two assumptions: 1) the solid behaves as an isotropic continuous medium as far as thermal vibrations are concerned and (2) all the deductions from the theory obtained for a monatomic substance automatically hold for a diatomic The present authors point out that both these assumptions are not strictly correct. In the majority of papers concerned with the X-ray measurement of @ as a function of temperature, attempts are made to obtain information on the corresponding change in the interatomic forces. analysis of experimental data has led the present authors to the as measured conclusion that the characteristic temperature . Card 1/4

S/139/60/000/006/015/032 E032/E414

Determination of the Characteristic (Debye) Temperature by X-Ray Methods

by the X-ray method, cannot be a simple characteristic of the interatomic forces in the lattice of a solid solution. the points quoted in support of this conclusion are: 1) small changes (of the order of 1 or 2% at.) in the concentration of the second component give rise to large changes in (20 to 30%) (V.A.Il'ina, V.K.Kritskaya, Ref.5); such a change in (a) cannot be due only to a change in the interatomic forces; 2) the characteristic temperature 📵 is irreversibly dependent on the heat treatment to which the alloys have been subjected and is not uniquely determined by the short-range order parameter on the first coordination sphere;
3) V.A.Il'ina et al (Ref.7) pointed out an analogy between and Young's modulus E in their dependence on heat treatment. △ @/@ and △E/E Analysis of numerical data for that the observed correspondence is only apparent, The debye theory predicts that  $\Theta \sim \sqrt{E}$  so that  $\Delta \Theta / \Theta = 1/2(\Delta E/E)$ . Experimental data, on the other hand, show that 🕲 changes by Card 2/4

S/139/60/000/006/015/032<sup>-</sup> E032/E414

Determination of the Characteristic (Debye) Temperature by  $X-Ray\ Methods$ 

20 to 30% while the corresponding change in E is less than 3 or 4%; 4) finally, V.I.Iveronova et al (Ref.8) and S.A.Nemnonov and Finkel'shteyn, L.D. (Ref.9) have shown that the measured values of are smaller in the ordered than in the unordered state (in the case of Cu3Au, Ni3Fe, Fe3Al). The present authors argue that the characteristic temperature as measured by X-ray methods, depends not only on the interatomic forces but also on other factors such as the spectrum of thermal lattice vibrations, the relation between the dynamic and static displacements, variation of the mean square static displacement user with temperature etc. It is concluded that no definite conclusions can be made about changes in the interatomic forces in the lattice on the basis of the X-ray measurements of alone. There are 1 figure and 15 references: 9 Soviet and 6 non-Soviet.

C\_rd 3/4

\$/139/60/000/006/015/032 E032/E414

Determination of the Characteristic (Debye) Temperature by X-Ray Methods

ASSOCIATION: Moskovskiy gosuniversitet imeni M.V.Lomonosov

(Moscow State University imeni M.V.Lomonosov)

SUBMITTED: July 11, 1960

Card 4/4

5.4700

S/181/60/002/01/25/035 B008/B014

24.7600 AUTHORS:

Zvyagina, A. P., Iveronova, V. I.

TITLE:

The Characteristic Temperature and the Spectrum of Thermal

Lattice Vibrations 1

PERIODICAL: Fizika tverdogo tela, 1960, Vol. 2, No. 1, pp. 118-121

TEXT: As a model of a solid solution the authors took a simple cubic lattice consisting of two atoms with the masses m<sub>1</sub> and m<sub>2</sub> at a concentration of 50 at%. In an absolutely disordered state the solution was thought of as a monatomic cubic lattice with an effective atomic mass

 $m=\frac{m_1+m_2}{2}$ . The model of an ordered solution represents an NaCl-type lattice. The spectra indicated in Ref. 5 for lattices of this type were utilized for the purpose, and  $\bar{u}^2$  (thermal lattice vibration frequency) was calculated for the disordered (Fig. 1) and ordered states (Fig. 2)

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The Characteristic Temperature and the Spectrum of Thermal Lattice Vibrations

S/181/60/002/01/25/035 B008/B014

at a mass ratio of  $\frac{m_1}{m_2} = 3$ . A quantitative comparison between theory and

experiment was impossible since the solid solution that was experimentally studied differed from the calculated model in its lattice and stoichiomentric composition. It was, however, established that the characteristic temperature  $\theta_{\rm r}$ , which was measured by the X-ray technique via  $\bar{\rm u}^2$ , cannot be regarded as an energy index of the intermolecular interaction in the solid solution.  $\bar{\rm u}^2$  depends not only on the cutoff frequency (binding energy) but also on the shape of the thermal vibration spectrum of the lattice. The latter is determined by such factors as lattice type, distribution of atoms in the lattice (degree of orientation), mass ratio of the components, and difference in binding energy between the individual atomic pairs. A close relationship between the change in  $\theta_{\rm r}$  and the

change in the binding energy can be established only in the simplest cases if the lattices have the same spectra, i.e., if the lattice type, the degree of orientation etc. do not change with varying concentration

Card 2/3

The Characteristic Temperature and the Spectrum of Thermal Lattice Vibrations

S/181/60/002/01/25/035 B008/B014

1. 1

en departura en en demonstrator en en describeradors de la constanción de la composition de la composition de Constanción de la composition de la comp

and thermal treatment of the solid solutions. Similar considerations hold for the characteristic temperature which was determined from the dependence of the heat capacity upon the temperature. This is why the heat capacity depends on the entire vibration spectrum, and not only on its fundamental frequency (Ref. 2). There are 2 figures and 9 references, 3 of which are Soviet.

ASSOCIATION: Moskovskiy gosudarstvennyy universitet (Moscow State University)

SUBMITTED: May 4, 1959

Card 3/3

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78101 24,7000 507/70-5-1-10/30

Iveronova, V. I., Katsnel'son, A. A. AUTHORS: The transmission of the section of t

Short-Range Order and Characteristic Temperature of TITLE:

NigPt Determined According to the X-Ray Data

Kristallografiya, 1960, Vol 5, Nr 1, pp 71-78 (USSE) PERIODICAL:

Earlier studies by the authors and others have disclosed ABSTRACT:

that short-range order parameter a, which charac-

terizes the varying number of B atoms on the first sphere around an A atom, is not proportional to the characteristic temperature  $\theta_{X-ray}$ , determined

& X-ray proved to according to the X-ray data.

increase during the initial brief period of annealing

of NI-Pt alloys and then to drop, while a having

reached certain level remained constant in the course of further annealing even for 20 hr. The authors have

now extended the study on the distribution of B atoms to the second sphere around A atoms and sought to

Card 1/7

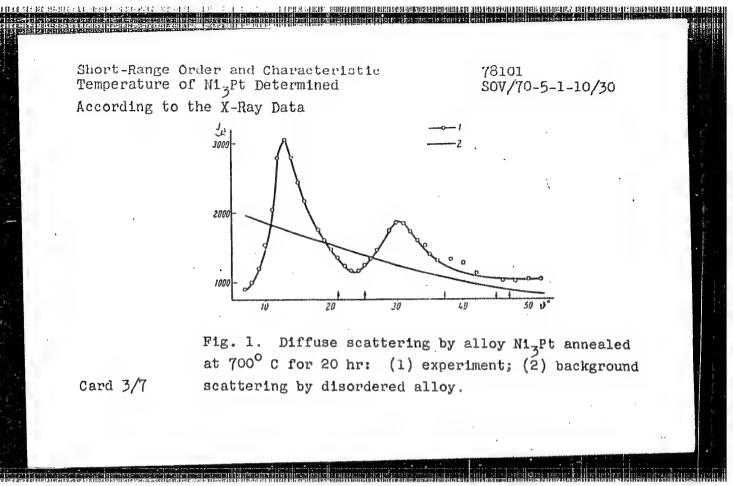
Short-Range Order and Characteristic Temperature of Ni3Pt Determined

78101 SOV/70-5-1-10/30

According to the X-Ray Data

determine  $a_1$  and  $a_2$  as functions of the mean square of dynamic atomic displacements at varying temperatures and durations of annealing. Night was obtained in a high-frequency furnace, homogenized at  $900^{\circ}\text{C}$  for 50 hr, checked for composition measuring identity period, cut into plates of  $16\times20\times2$  mm, deformed plastically by rough grinding, annealed at 700°C to 1.050°C for 20 hr under vacuum better than 0.001 mm, and quenched in water to fix the achieved state. The X-ray diffraction photographs were taken at 20 and -160°C with Mo radiation; the computations were based on diffuse scattering data (Fig. 1) at glancing angles (10) from 7 to  $42^{\circ}$ , within which two maxima corresponded to  $a_1$  and  $a_2$ . The increase of annealing temperature from 700 to 1,050°C proved slowly to

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Short-Range Order and Characteristic Temperature of Ni<sub>3</sub>Pt Determined According to the X-Ray Data 78101 SOV/70-5-1-10/30

reduce  $a_1$  and increase  $a_2$ . The improved order in the structure after annealing is more precisely defined by the sum  $\Delta$   $\frac{PtNi}{n_1}$  +  $\Delta$   $\frac{PtNi}{n_2}$ , the terms of

which, respectively, denote the numbers of Ni atoms on the first and second spheres around Pt in excess of the respective numbers before annealing (disordered state). The first term reaches a maximum within an

hour of annealing at 700°C and then remains constant, while the second term (a negative value) continues to rise even after 20 hr. Annealing at 1,000°C establishes the second term also at a maximum, pointing to the restoration of order and equilibrium in the deformed alloy within both spheres. The higher the temperature of annealing, the sum approaches zero in both cases. At lower temperatures the sum shows a sharp maximum at brief annealing.

Card 4/7

Short-Range Order and Characteristic Temperature of Ni<sub>3</sub>Pt Determined According to the X-Ray Data

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This obviously is an effect of bond energy  $W_2$  which forces N1 atoms to move from the third sphere and to concentrate on the second, while greater atomic mobility at a higher temperature provides for rapid motion of N1 atoms into the first sphere at the expense of the second, where consequently no N1 concentration takes place. The first term becomes sooner stabilized at a constant value since the constant in this case means the establishment of equilibrium between only two spheres, while the second term involves the equilibrium between first, second, and third spheres. The table below demonstrates that characteristic

temperature  $\Theta$  changes proportionally to the X-ray above sum. The sixth column shows mean squares of dynamic atomic displacements whose increase with the temperature and duration of annealing reduces the sum. Thus, a cannot define characteristic temperature alone

Card 5/7

Short-Range Order and Characteristic Temperature of Ni3Pt Determined

78101 SOV/70-5-1-10/30

According to the X-Ray Data

since both  $a_1$  and  $a_2$  affect it, while the sum

△ PtNi + △ PtNi defines it accurately. There

are 4 figures; 1 table; and 12 references, 10 Soviet,

1 Danish, 1 U.S. The U.S. is: B. E. Warren, B. L. Aberbach, B. W. Roberts, J. Appl. Phys, 22,

1493-1496 (1951).

Moscow State University imeni M. V. Lomonosov ASSOCIATION:

(Moskovskiy gosudarstvennyy universitet imeni M. V.

Lomonosova)

SUBMITTED:

July 2, 1959

Card 6/7

Short-Range Order and Characteristic Temperature of Ni<sub>3</sub>Pt Determined According to the X-Ray Data

78101 SOV/70-5-1-10/30

Table

T. °C	ubr.	An PUNI	Δn <sup>PtNI</sup>	$\Delta n_{1}^{\mathrm{PtNi}} + \Delta n_{2}^{\mathrm{PtNi}}$	u² (A²)	0° K
700 800 900 1050 defor 700 700 700	20 20 20 20 20 20 4 20 50	1,4 1,4 1,3 <sub>5</sub> 1,1 <sub>5</sub> 0,6 1,4 1,4	-1,1 -1,1 -1,1 <sub>3</sub> -1,3 -0,7 -0,9 -1,1 -1,1	0,3 0,3 0,2 -0,1 <sub>s</sub> -0,1 0,5 0,3 0,3	0,011 	270 280 410 330

Card 7/7

IVERONOVA, V.I.; POPOVA, I.I.; REVKEVICH, G.P.

Effect of faults in the packing of layers on the intensity of Debye lines. Kristallografiia 5 no.4:530-534 Jl-Ag '60. (MIRA 1319)

1. Moskovskiy gosudarstvennyy universitet im. M.V. Lomonosova. (Brass-Spectra)

\$/070/60/005/005/024/026/XX E132/E160

AUTHORS:

Iveronova, V.I., and Katsnel'son, A.A.

TITLE:

The Methodology of Measuring the Intensity of the Diffuse

Scattering from Polycrystalline Materials

PERIODICAL: Kristallografiya, 1960, Vol.5, No.5, pp. 795-797

Measurement of the diffuse scattering from powders is more difficult than from single crystals. A method must have three elements: a means for excluding the parasitic components of the background (white radiation, fluorescence, Kp lines, harmonics of the lines, scattering by materials other than the specimen); a means of lines, scattering by materials other than the specimen); measuring the background with at worst 2-3% error; the possibility of putting background intensity measurements on an absolute (electron A technique has been worked out for measuring diffuse scattering using a YP( -50 W (URS-501) diffractometer without employing a vacuum camera. Radiation incident on the specimen is monochromatized by reflexion (002 plane) from a crystal of pentaerithritol where harmonics (004, 006 intensities) are very low. A counter with some discrimination against unwanted wavelengths is used. Specimens were pressed into a plate 18 x 14 x 2 mm without binder. Air scatter was reduced with a special collimating system Card 1/2

S/070/60/005/005/024/026/XX B132/E160

The Methodology of Measuring the Intensity of the Diffuse Scattering from Polycrystalline Materials

and the residual scatter could be estimated from tests without a specimen. Estimates of the efficiency of the general arrangement were made by working with and without collimator and with and without monochromator and measuring the ratio of line intensity to background intensity. A specimen of Ni3Pt was studied with Fe radiation after checking the measured diffuse scattering from Ni with the theoretical values. The agreement is very satisfactory and confirms the accuracy and utility of the apparatus. There are 3 figures and 7 referencess 4 Soviet and 3 English.

ASSOCIATION: Moskovskiy gosudarstvennyy universitet im.

M.V. Lomonosova

(Moscow State University Imeni M.V. Lomonosow)

SUBMITTED: March 13, 1960

Card 2/2

S/126/60/010/005/016/030 E193/E483

AUTHORS: V.I. and Osipenko, N.N.

TITLE: Recrystallization of Pure Metal Powders

PERIODICAL: Fizika metallov i metallovedeniye, 1960, Vol.10, No.5,

pp.736-742

In many metallographic investigations, in which X-ray diffraction technique is used, specimens characterized by random distribution of crystals regarding their orientation have to be employed and this necessitates the use of powder specimens These are used either in the deformed or in the In the former case, it is usually assumed (filings). that the internal stresses in the powder particles are larger than those present in a massive specimen, even more heavily deformed. However, problems such as what is the structure of powder specimens produced by filing, to what degree they have been deformed, and what is their recrystallization temperature, have not been systematically studied, although there are indications that recovery processes can take place in filings even at room temperature and that their recrystallization temperature is higher than that of If accurate deductions, heavily deformed massive metal specimens. Card 1./4

S/126/60/010/005/016/030 E193/E483

Recrystallization of Pure Metal Powders

regarding the structure of plastically deformed massive metal, are to be made from experimental results obtained on deformed powder particles, it is necessary to know the effect of room temperature ageing and high-temperature annealing on the properties of powder specimens. The object of the present investigation was to study the structural changes in copper and aluminium filings: (a) aged for various periods at room temperature in the case of copper and at 60 and 100°C in the case of aluminium and (b) annealed for a given time at various To avoid the effects of heat produced by friction, the experimental powder samples were prepared by slow filing. Four samples of each metal were prepared; samples a and b from fully annealed massive specimens, and samples c and d from heavily deformed specimens; each sample was separated into the fine (samples a and c) and coarse (samples b and d) fractions. After being subjected to various heat treatments, the powders were examined by X-ray diffraction. From the variation of the number of spots on the X-ray diffraction pattern, deductions were made Card 2/4

S/126/60/010/005/016/030 E193/E483

Recrystallization of Pure Metal Powders

ragarding the variation of the proportion of non-distorted crystals in the specimen and the temperature interval of the recrystallization process; the magnitude of the stresses of the second type, and the dimensions of the mosaic blocks in the specimens, were determined from broadening of the X-ray diffraction The results indicated that in filings annealed even at comparatively low temperature (at room temperature in the case of copper), polygonization takes place, which leads to a decrease in the average size of the mosaic blocks, relief of the stresses of the second type and appearance of spots (due to reflections from undistorted crystals) on the X-ray pattern. This process takes place more readily in an isolated grain and is inhibited if slipping is hindered by forces exerted by the adjacent crystals; it was for this reason that the intensity of this process was higher in fine powder samples, prepared from annealed materials. Recrystallization proper begins in metal powders only at temperatures near or above 500°C. Even then, polygonization takes place in the initial stages of the process, as a result of Card 3/4

GRAYEVSKAYA, Ya.I., IVERONOVA, V.I.; TARASOVA, V.P.

Effect of the concentration of tin in Cu-Sn solid solutions on the determination of characteristic temperature by X ray. Vest. Mosk. un. Ser. 3: Fiz., astron. 15 no. 4:52-58 '60. (MIRA 13:9)

 Kafedra obshchey fiziki dlya fizikov Moskovskogo universiteta. (Copper-tin alloys-Thermal properties) (Heat capacity)

KRYLOV, Ya.; IVERONOVA, V.I.

Flastic deformation of germanium single crystals (by insertion of a spike). Kristallografiia 6 no.5:784-786 S-0 '61.

(MIRA 14:10)

1. Moskovskiy gosudarstvemnyy universitet imeni Lomonosova.

(Deformations (Mechanics)) (Germanium crystals)

5/126/61/011/001/004/019 E032/E314 Iveronova, V. I. and Katsmel'son, A.A. **AUTHORS:** The Dimensional Effect in X-ray Diffraction by TITLE: Polycrystalline Materials Fizika metallov i metallovedeniye, 1961, PERIODICAL: Vol. 11, No. 1, pp. 40 - 45 Diffuse scattering by polycrystalline alloys, whose TEXT: atoms are displaced from their ideal positions in the lattice, has been treated by Warren et al (Ref. 1) and Gouska and Averbach (Ref. 2). However, these authors assumed that the magnitude of the displacements depends only on the type of the atoms under consideration and not on the type of their nearest neighbours. Krivoglaz (Ref. 3) has also treated this problem without the introduction of the above restrictions. However, the latter author was concerned only with the diffuse background in the neighbourhood of reflections. The present authors extend the theory of diffuse scattering by polycrystalline materials to any angles of reflection. discussion is based on the formula: Card 1/4

S/126/61/011/001/004/019 E032/E314

The Dimensional Effect in X-ray Diffraction by Polycrystalline

$$I_{ii} = Nc_{A} c_{B} \left[ \sum_{l=0}^{\infty} \sum_{l=0}^{c_{l}} \alpha_{il} \cos \vec{k} \vec{R}_{il} \right] e^{-2\pi_{\text{er}} \left( \frac{4 \ln \delta}{\lambda} \right)^{2}} \left[ \int_{B} -f_{A} + \frac{a_{g} \vec{f} \vec{k} \vec{e_{g}}}{|\vec{g}|} \right]^{2}, \quad (1)$$

which was derived by Krivoglaz in Ref. 4 for the case of diffuse scattering by a monocrystal and is free from the limitations introduced in Refs. 1 and 2. This formula consists of two parts, the first of which is due to the difference in the scattering power of the components of the alloy and their atomic radii and the second part represents positions in a perfect lattice, i.e. the so-called Card 2/4

### S/126/61/011/001/004/019 E032/E314

The Dimensional Effect in X-ray Diffraction by Polycrystalline Materials

dimensional effect. Expressions are derived for the intensities of diffusely scattered X-rays associated with the above two effects. The intensity associated with the first of these effects is found to be a quasiperiodic function of the angle of reflection and is in qualitative agreement with the results of Warren et al (Ref. 1). In the case of the second effect the intensity increases with angle. It is found that the positions of the maxima and minima in these distributions depend on the type of crystal lattice, the scattering power and radii of the atoms of the components of the alloy as well as on the magnitude of the short-range order parameters. There are 3 figures and 7 references: 3 Soviet and 3 non-Soviet.

Card 3/4

S/126/61/011/001/004/019
E032/E314

The DimensionalEffect ....

ASSOCIATION: Moskovskiy gosudarstvennyy universitet im.
M.V. Lomonosova (Moscow State University im.
N.V. Lomonosov)

SUBMITTED: June 24, 1960

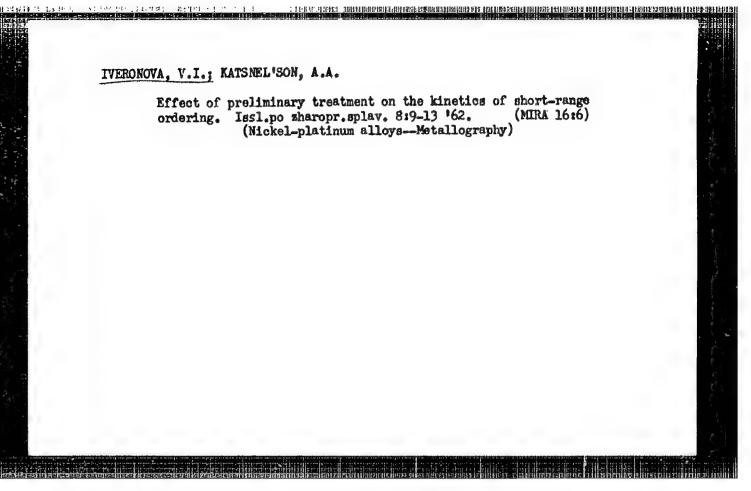
Card 4/4

APPROVED FOR RELEASE. US/20/2002 TVERONOVA, V.I.; KATSNEL'SON, A.A. Short-range order and physical properties of single-phase solid solutions (survey). Zav.lab. 27 no.11:1354-1361 161. (MIRA 14:10) 1. Moskovskiy gosudarstvennyy universitet imeni M.V. Lomonosova. (Solutions, Solid)

BELYANKIN, A.G.; MOTULEVICH, G.P.; CHETVERIKOVA, Ye.S.; YAKOVLEV,
I.A.; IVERONOVA, V.N., tekhn. red.

[Laboratory manual on physics] Fizicheskii praktikum. Pod
red. V.I.Ivernovpi. Moskva, Fizmatgiz, 1962. 956 p.

(Physics—Laboratory manuals)



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IVERONOVA, V.I.; KATSNEL'SON, A.A.

Determination of the characteristic temperature based on the diffuse scattering of polycrystals. Kristallografiia 7 no.4: 616-618 J1-Ag '62. (MIRA 15:11)

1. Moskovskiy gosudarstvennyy universitet imeni Lomonosova. (Crystals--Thermal properties)

IVERONOVA, V.I.; KATSNEL'SON, A.A.

Effect of short range order on the hardness and characteristic temperature of Ni<sub>3</sub>Pt. Fiz. met. i metalloved. 13 no.2:317-319 F '62. (MIRA 15:3)

 Moskovskiy gosudarstvennyy universitet im. M.V.Lomonosova. (Nickel-platinum alloys-Thermal properties)

IVERONOVA, V.I.; ZVYAGINA, A.P.; AYNBINDER, B.Yu.

Effect of the spectral line shape of thermal vibrations in a CaCl-type lattice on heat capacity, mean square shift of atoms from the state of equilibrium, and the speed of sound.

Fiz. met. i metalloved. 14 no.1:141-144 J1 162. (MIRA 15:7)

1. Moskovskiy gosudarstvennyy universitet imeni Lomonosova.
(Crystal lattices) (Spectrum, Atomic)

\$/048/62/026/003/002/015 B107/B102

AUTHORS:

Zvyagina, A. P., and Iveronova, Y. I.

TITLE:

Spectrum of thermal vibrations and the characteristic

temperature of a CsCl-type lattice

PERIODICAL: Akademiya nauk SSSR. Izvestiya. Seriya fizicheskaya, v. 26,

no. 3, 1962, 340-344

TEXT: The mean square displacement of the atoms from equilibrium, is

 $\overline{u^2} = \frac{1}{6N^2} \int_0^{\nu_m} u^2(\nu) g(\nu) d\nu = \frac{A}{6N^3} \int_0^{\nu_m} \frac{1}{\nu} \left( \frac{1}{e^{\hbar \nu/kT} - 1} + \frac{1}{2} \right) g(\nu) d\nu,$ 

where g(v)dv is the number of frequencies in the interval between v and where g(v) is the number of frequencies in the interval between v and v+dv, g(v) is the distribution function of the frequencies, i.e. the spectrum of thermal vibrations,  $2N^3$  is the number of atoms in the crystal,  $v_m$  is the maximum frequency. According to Debye,  $g(v) = cv^2$ , but the real spectrum shows considerable deviations from Debye's parabolic law, Card 1/4

Spectrum of thermal vibrations ...

S/048/62/026/003/002/015 B107/B102

especially for diatomic crystals. The spectrum of thermal vibrations was calculated for a CsCl-type body-centered cubic lattice by Montroll's method (Refs. 2, 3, see below). Central forces were assumed and interaction of atoms with the nearest and next nearest neighbors was taken into consideration. The spectrum is represented by expansion into Legendre polynomials:

$$g(v_m x) = a_0 + a_2 P_2(x) + a_4 P_4(x) + \dots + a_{14} P_{14}(x),$$

where  $x = v/v_m$ ,  $0 \leqslant x \leqslant 1$ . The coefficients  $a_{2k}$  are expressed in terms of the even moments of the distribution function

 $\mu_{2k} = \frac{1}{6N^3} \int_0^M v^{2k} g(v) dv$ . The moment  $\mu_{2k}$  is calculated as the trace of the

characteristic matrix raised to the k-th power. The matrix elements depend on the ratio of the four parameters

$$a = \frac{2\alpha_{18}}{3\pi^2 m_1 v_m^2}, \quad b = \frac{2\alpha_{18}}{3\pi^2 m_2 v_m^2}, \quad c = \frac{\beta_{11}}{\pi^2 m_1 v_m^2}, \quad d = \frac{\beta_{21}}{\pi^2 m_2 v_m^2},$$

where  $m_1$  and  $m_2$  are the masses of the two types of atoms;  $\alpha_{\mbox{i}\mbox{j}}$  and  $\beta_{\mbox{i}\mbox{j}}$  are Card 2/5

Spectrum of thermal vibrations ...

S/048/62/026/003/002/015 B107/B102

the coefficients of quasielastic interaction of the i-th and j-th types of atoms in the first and second sphere of coordination, respectively. The even moments of the distribution function are given explicitely. For equal atomic masses (a=b), but different coefficients of quasielastic interaction, the shape of the spectrum is given by the parameters

$$\gamma = \frac{c}{a} = \frac{33_{11}}{2\alpha_{12}}$$
 is  $\delta = \frac{d}{a} = \frac{3\beta_{22}}{2\alpha_{12}}$ ,

which are within -0.2 and +0.2. Fig. 1 shows the spectra for  $\gamma = -0.2$  and various  $\delta$  values. Breaking off the series had a great effect and did not permit the spectra to be calculated for different atomic masses. The

quantity  $\Delta u^2 = u_R^2 - u_N^2$  which can be determined experimentally was calculated from the spectra;  $u_R^2$  and  $u_N^2$  are the mean square displacement of the atoms at room temperature and at liquid-nitrogen temperature,  $\Delta u_N^2$  are the mean of temperature,  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the mean square displacement of  $\Delta u_N^2$  and  $\Delta u_N^2$  are the

 $\Delta u^2$  on  $\delta$  at various  $\gamma$  values. The lower families of curves hold for Card 3/5

Spectrum of thermal vibrations ...

S/048/62/026/003/002/015 B107/B102

equal ratios of the binding parameters, but for an absolute value increased to 4/3 and 2, respectively. The authors thank B. Yu. Aynbinder for assistance. There are 2 figures. The two most important English-language references are: Ref. 2: E. W. Montroll, J. Chem. Phys., 11, 481 (1943); Ref. 3: E. W. Montroll, D. C. Peaslee, J. Chem. Phys., 12, 98 (1944).

ASSOCIATION: Moskovskiy gos. universitet.im. M. V.Lömonosova (Moscow State University imeni M. V. Lomonosova (Moscow

Fig. 1: Spectra of thermal vibrations for CsCl-type lattices with the binding parameter  $\gamma$  = -0.2.

Fig. 2: Dependence of  $\Delta u^2$  on  $\gamma$  and  $\delta$ .

Card 4/5

S/185/63/008/002/009/012 D234/1508

AUTHORS:

Iveronova, V. I. and Zvyagina, A. P.

TITLE:

Phonon spectrum and some thermodynamical characteris tics of CaCl type lattices

PERIODICAL:

Ukrayins kyy fizychnyy zhurnal, w. 8, no. 2, 1963,

TEXT: The authors refer to a previous paper of theirs (FMM, v. 14, 141, 1962) and conclude that v/a and other elastic constants can

be used as direct characteristics of interactions of atoms, while  $C_{v}$  and  $\Delta_{u}^{-2}$ cannot generally be so used within the limits of Debye's theory. An example connected with the computation of characteristic temperatures is given. The authors also indicate a graphical method for estimating the interaction parameters from the above quantities. There are 5 figures.

ASSOCIATION:

Moskovskiy gosuniversitet im. M. V. Lomonosova (Moscow State University im. M. V. Lomonosov)

Card 1/1

AUTHOR: Iveronova, V. I.; Katsnel'son, A. A.

TITLE: Diffuse and double Bragg scattering of polyerystalline double

SOURCE: Kristallografiya, v. 8, no. 3, 1963, 463-465

TOPIC TAGS: scattering, Compton effect, thermal scattering, Cu, secondary extinction

ABSTRACT: Electrolytic copper in cakes of powder pressed at room temperature and pressure of 1-2 tons/Sq cm, in cakes pressed at 8000 and pressure of 60 tons/Sq cm, and in sintered samples was used. The results show that scattering from powdered samples, after long heating at 8000, was due to thermal and Compton effects. For sintered samples and powder samples annealed at 4000, subsidiary treatment provided double Bragg scattering. The authors discovered a relation between double Bragg intergranular scattering and secondary extinction. They concluded that double scattering was missing in the pressed cakes because of development of crystal units in the grains, each grain becoming essentially a single crystal/f. The total number of units in the zone of scattering thus becomes small and conditions for double scattering disappear. In the sintered samples this situation does not arise, and double scattering is present. It is further concluded that diffuse scattering by Cord 1/2

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(Moscow Sta	ate University)	- warstvennymy w	niversitet im. M. \	/. Lomonosova	
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S/126/63/015/002/015/033 E039/E420

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AUTHORS: Vlasova, Ye.N., Iveronova, V.I.

TITLE: The diffuse scattering of X-ray bonns and the real

structure of equilibrium solid solutions

PERIODICAL: Fizika metallov i metallovedeniye, v.15, no.2, 1965,

254-259

TEXT: A study was carried out of the distribution of intensity of diffuse scattering of an X-ray beam in the reciprocal lattice of annealed Fe-Al alloy with 15 at. % Al. The analysis leads to the conclusion that the distribution of atoms of iron and aluminium at the nodes of the crystal lattice does not appear by chance. suggestions are made relating to the structure of a solid solution: (1) that there is an orderly arrangement of atoms and that the order is not less than in three spheres of coordination and (2) that the structure of the solid solution is a non-predered matrix with a statistical distribution of concentrations of Al atoms in its zones. These zones have an ordered structure of the type Fe3Al. Comparison of experimental data with the theoretical conclusions of Borie shows that the basic facts such as the displacement of the maximum near the even modes of the Card 1/2

The diffuse scattering ...

S/126/63/015/002/015/033 E039/E420

superstructure at small angles, the invariable positions and intensity maxima for odd superstructure nodes etc can be explained by the distortion of the crystal lattice as a result of the large difference in size of the atoms of Al and Fe. The results are compared with diffuse scattering in the alloy Cughu. There are 3 figures.

ASSOCIATION: Institut pretsizionnykh splavov TsNIIchM

(Institute of Precision Alloys TsNII(thM)

SUBMITTED: July 17, 1962

Card 2/2

S/057/63/033/001/014/017 B125/B186

AUTHORS:

Iveronova, V. I., Kagan, A. S., and Nikolayeva, S. M.

TITLE:

On the measurement of the relative intensities of X-ray photographs of samples with texture

PERIODICAL: Zhurnal tekhnicheskoy fiziki, v. 33, no. 1, 1963, 115 - 117

TEXT: A general method is suggested for determining the relative intensities of the second orders of reflection in X-ray photographs of samples with textures. A method is sought of liminating any effect that a texture may exert on the relative intensity of second-order: reflections. In this case, either the region of the pole figure that falls in the reflecting position must be uniform, i. e. the slit heights must be different for different orders of reflection so that  $h_{\rm kl}/h_{\rm 2k2k2l} = \frac{1}{2}$ , or it must be ensured beforehand that the intensity of reflection inside the slit height does not change in the first order. The angle factor in the formula for the theoretical intensity is to be calculated according to the formula  $J \approx (1 + \cos^2 2\theta)/\sin 2\theta$ . Different slit heights must be Card 1/2

On the measurement of ...

S/057/63/033/001/C14 /017 B125/B186

used according to whether axial textures are photographed by the Bragg-Brentano or by the Debye method. The consideration of the nonuniform density of the normals occasions only a negligible error in determining intensity from the slit height. For calculating the error in photographing with the same slit heights the actual distribution of the normals must be known in each case. Differences  $\Delta(J_{2n2k21}/J_{nk1})$  lead throughout to noticeable statistical fluctuations having the order of magnitude of d/30. There are 2 figures.

ASSOCIATION: Nauchno-issledovatel'skiy konstruktorsko-tekhnologicheskiy institut podshipnikovoy promyshlennosti, Zagorsk (Scientific Research Design-technological Institute of the Bearing Industry, Zagorsk)

SUBMITTED: February 12, 1962(initially)
April 9, 1962 (after revision)

Card 2/2

IVERONOVA, V.I., prof., red.; GRABOVSKIY, M.A., dots., red.;
KONONKOV, A.F., kand. fiz.-mate. nauk, red.; MALOV, N.N.,
prof., red.; TELESNIN, R.V., prof., red.; USAGIN, S.I.,
st. prepod., red.; YAKOVLEV, K.P., prof., red.; YAKOVLEV,
I.A., prof., red.

[Methodology and technique of lecture demonstrations in physics; transactions] Metodika i tekhnika lektsionmykh demonstratsii po fizike; sbornik trudov. Moskva, Izd-vo Mosk. univ., 1964. 280 p. (MIR. 17:5)

1. Mezhvuzovskaya konferentsiya po lektsionnym demonstratsiyam po kursu obshchey fiziki. lst.

ACCESSION NR: APLO117LL

\$/0181/64/006/001/0101/0107

AUTHORS: Belov, K. P.; Iveronova, V. I.; Zaytseva, M. A.; Kadomtseva, A. M.; Katsnel'son, A. A.; Yatskul'yak, K.

TITLE: Magnetic and structural properties of lanthanum orthoferrite during partial replacement of Fe 3\* ions by other trivalent ions

SQURCE: Fizika tverdogo tela, v. 6, no. 1, 1964, 101-107

TOPIC TAGS: magnetic property, structural property, orthoferrite, lanthanum, lanthanum orthoferrite, Fe 3+, Al 3+, Sc 3+, Co 3+, thermoremanent magnetization, magnetization intensity, hysteresis loop, crystal lattice

ABSTRACT: In these studies the Fe<sup>+3</sup> ion was replaced, in part, by Al<sup>+3</sup>, Sc<sup>+3</sup>, Cr<sup>+3</sup>, and Co<sup>+3</sup>. Thermoremanent magnetization of LaFeO<sub>3</sub> cannot be reduced to zero even in a field of 20 000 cersteds, but if Al<sup>+3</sup> ions replace some of the Fe<sup>+3</sup> ions (InFe<sub>O.9</sub> Al<sub>O.1</sub>O<sub>3</sub>), introduced by orthorhombic distortion of the crystal lattice, thermoremanent magnetization almost disappears, and the hysteresis loops become symmetrical. These changes may be explained by the finely dispersed character of the samples. The change in magnetic properties on substitution of the indicated ions Card 1/2

ACCESSION NR: APLOLITAL

is associated with change in degree of dispersion and with the orthorhombic distortion of the lattice. Along with these changes, an increase was observed in magnetization intensity. This is explained by the ordered distribution of Al®3 ions in the crystal lattice. Orig. art. has: 3 figures.

ASSOCIATION: Moskovskiy gosudarstvenny\*y universitet im. M. V. Lomonosova (Moscow State University)

SUBMITTED: loJul63 DATE ACQ: lliFeb6l ENCL: OO

SUB CODE: FH NO REF SOV: OO2 OTHER: OO2

GLAGOIEVA, V.P.; IVERONOVA, V.I.; KASSANDROVA, U.N.

Effect of the K-state on the extent of the root-mean-square shifts of atoms in a Fe-Al alloy. Inv. vys. ucheb. zav.; (MIRA 17:11)

1. Moskovskiy gosudarstvennyy universitet imeni Lomonosova.

GRABOVSKIY, Mikhail Aleksandrovich, dots.; MLODZEYEVSKIY, Anatoliy
Boleslavovich, prof.; TELEGIIN, Roman Vladimirovich, prof.;
SHASKOL'SKAYA, Marianna Fetrovna, dots.; YAKOVLEV, Ivan
Alekseyevich, prof.; IVERONOVA, V.I., red.; CHEBOTAREVA,
A.V., red.

[Lecture demonstrations in physics] Lektsionaye demonstratsii po fizike. Moskva, Nauka, 1965. 572 p.

1. Institut stali i splavov Moskva (for Shaskol'skaya).

IVERONOVA, V.I.; OSIPENKO, N.N.

Low-temperature annealing of plastically deformed metals.
Fiz. met. i metalloved. 20 no.3:417-423 S '65.

(MIRA 18:11)

1. Moskovskiy gosudarstvennyy universitet imeni M.V.

Lomonosova.

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L 15756-66 ENT(1)/ENT(m)/T/EMP(t)/ENP(b) JD/LHB  ACC NR: AP5027459 SOURCE CODE: UR/OC32/65/031/011/1349/1352
Calpenko, H. H.
universitet) . H. V. Iomonosov, Moscow (Muskovskiy gosudarstyspary
TITLE: Determination of the size of mosaic blooks and michoscopic lattice
SOURCE: Zavodskaya laboratoriya, v. 31. no. 11. 706s 12.00
distribution, cauchy distribution, harmonic analysis, calculation, gaussian
ABSTRACT: A method is proposed for determining the parameters of the fine structure
plasticity. Two Debye lines (designated A and B) with imisctropy of the modulus of
does not depend on the indexes (bkl) in the direction studied, and (2) the stresses

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ACC NR: AP5027459

During calculations by the Gauss approximation, it is necessary to replace

 $\frac{1}{\Omega}$ ,  $4\frac{6a}{a}$ ,  $\beta\cos\theta$ ,  $\beta\sin\theta$ 

and E in the above formulas with the expressions

 $\left(\frac{\lambda}{A}\right)^2$ ,  $4\left(\frac{\delta \alpha}{\alpha}\right)^2$ ,  $\beta^2 \cos^2 \theta$ ,  $\beta^2 \sin^2 \theta$ .

and £2, correspondingly. These methods of calculation were checked experimentally with nickel, copper, and armoo iron. Drig. art. has: 3 figures and 9 formulas.

SUB CODE: 20,12/ ORIG REF: 005/ OTH REF: 000

10

Card 3/3 10

L h6283-66 ENT(m)/EMP(w)/T/EMP(t)/ETI/EMP(k) IJP(c) JD/HW

ACC NR. AP5025327 SOURCE CODE: UR/0126/65/020/003/0417/0423

AUTHOR: Iveronova, V. I.; Osipenko, N. N.

ORG: Moscow State University im. M. V. Lomonosov (Moskovskiy gosuniversitet)

TITLE: Low temperature annealing of plastically deformed metals

SOURCE: Fizika metallov i metallovedeniye, v. 20, no. 3, 1965, 417-423

TOPIC TAGS: plastic deformation, metal heat treatment, copper, powder metal, low temperature effect, powder metal property, metal deformation, annealing at temperatures to 105C under iso-

ABSTRACT: The change of block structure by annealing at temperatures to 105C under isothermal conditions and/or by up to 6-month storage at room temperature was studied with compact copper samples, deformed up to 82% by rolling. Low, prerecrystallization temperatures or storage cause a decrease and a subsequent increase in the size of structural microblocks with simultaneous changes in microhardness. The starting size of blocks, derived from x-ray interference measurements, is higher than after storage at low temperature, and some stability in block size, microhardness and microdeformations is reached after sufficient storage at room or slightly elevated temperature. Annealing times, required for minimum size of block structure, and time required for stabilization of block size and microhardness

Card 1/2

UDC: 621,785,3

ACC NR. AP. 005327

SOURCE CODE: UR/0131/66/003/632/3459/3462

AUTHOR: Iveronova, V. I.; Tikhonov, A. N.; Zaikin, P. N.; Zvyagina, A. P.

ORG: Moscow State University im. M. V. Lomonosov (Moskovskiy gosudarstvennyy universitet)

TITLE: Determination of the phonon spectrum of crystals from the specific heat

SOURCE: Fizika tverdogo tela, v. 3, no. 12, 1966, 3459-3462

TOPIC TAGS: phonon spectrum, distribution function, specific heat, crystal property, thermodynamic function, aluminum

ABSTRACT: By using an approximate relation between the frequency distribution function and the specific heat, the authors demonstrate that in the harmonic approximation it is possible to calculate the phonon spectrum of crystals from the specific heat and from other thermodynamic functions. The approximate frequency distribution function is obtained directly from the experimental data on the specific heat. The determination of the approximate distribution function is facilitated by the fact that, in the approximation considered, the phonon spectrum is a continuous and piecewise smooth function with a derivative haveing a finite number of discontinuities. The resultant approximation is a smooth function which carries a minimum of characteristic information (line structure) and satisfies the equation with a specified accuracy. By way of an example, the frequency distribution function of aluminum, obtained from the integral equation using experimental information on the specific heat of aluminum,

Card 1/2

ACC NR: AP7005827

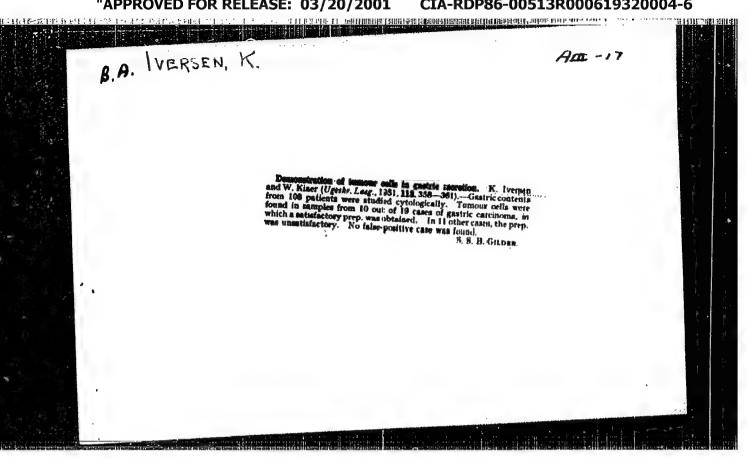
As presented. It is noted in the conclusion that the method can be used without modification to determine the energy spectrum of any Bose system from its thermodynamic functions. Orig. art. has: 2 figures and 9 formulas.

SUB CODE: 20/ SUEM DATE: 03Jan66/ ORIG REF: 005/ OTH REF: 003

IVEFONOVA, V.I.; KATSNEL'SON, A.A.

Existence of a short-range order in Au-Pd alloys. Kristallografiia 9 no.4:557-558 '64. (MIRA 17:11)

1. Moskovskiy gosudarstvennyy universitet imeni Lomonosova.



Sounds 000s: 00/0413/00/000/019/0031/0031 INVENTOR: Bliznyuk, N. K.; Kolomiyets, A. F.; Ivershina, L. P. ORG: none TITLE: Preparation of phosphonic ester chlorides. Class 12, No.186466 Izobreteniya, promyshlennyye obraztsy, tovarnyye znaki, no. 19. SOURCE: 1966. 31 TOPIC TAGS: phosphonic acid, ester, chloride, ainst phosphinate, phosphorus to chloride, chlorinatel organic compound, organic phosphorus aunquient ABSTRACT: To broaden the raw material base for the preparation of phosphonic ester chlorides from monoalkyl phosphinates and a chlorinating agent, a mixture of phosphorus trichloride and chlorine is used as the chlorination agent. [PS] [WA-50; CBE No. 14] SUB CODE: 07/ SUBM DATE: 04Sep64 Cardl /1 UDC: 547.26' 118.07

USSR/Diseases of Farm initrals. Diseases Caused by Helminths

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.bs Jour : Ref Zhur - Biol., No 19, 1958, No 83266

Luthor

Inst

: Boyev S.N., Ivershine Yo.M. : Institute of Veterinchy Medicine of the Kazakh Branch of the All-Union Academy of Agricultural Sciences inemi Lemin

Titla

: Spread and Dynamics of Ovine Hemo-Onchocorcosis in Kazalih-

Orig Pub : Tr. In-te vet. Kezelchek. fil. V.SKhNIL, 1997, 8, 404-416

Abstract : No abstract.

Card : 1/1

22

je. m. R : ÜSSR COUPTRY : Diseases of Farm Animals. Diseases Caused CATEGORY by Helminths : RZhBiol., No. 6 1959, No. 25995 ABS. JOUR. : Boyev, S. H.; Ivershina, Ye. H. : Institute of Zoology, AS Razssa AUTHOR INST. : On the Propagation and Dynamics of Intestinal Costodiases of Cattle in Kazakhstan TITLE ORIC. PUB. : Tr. In-ta zool. AN KazSSR, 1958, 9, 10-18 : As a result of the study of veterinary statistical data, it was shown that the loss of cattle ABSTRACT due to intestinal costodiases was recorded during the last 16 years in 11 out of 16 oblasts of Kazakhstan. Intestinal cestodiases of sheep are recorded in all oblasts of Kazakhstan in all seasons of the year. In the majority of the oblasts of the Republic, the nurrain of sheep from 1/2 CALD:

Osnovnyye printsipy planirovaniya I raspredeleniya dokhodov zheleznykh dorog SSR (Fundamental principles of planning and allocation of revenue of railroads in the USSR) Moskva, Transzheldorizdat, 1951.

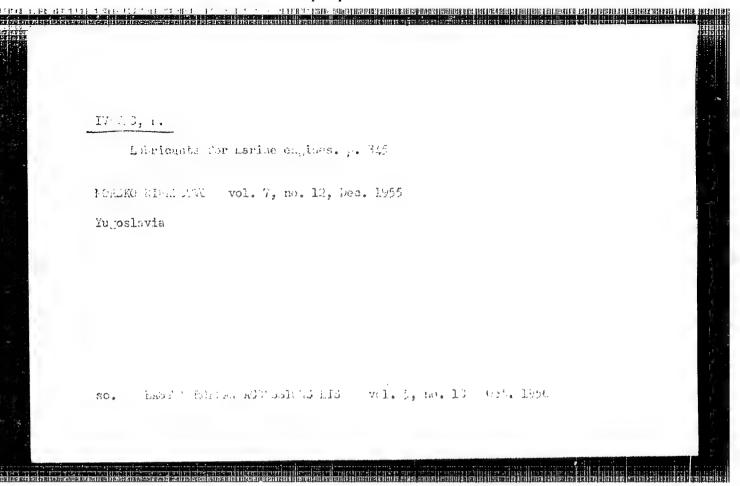
156 p. diagra., tables.

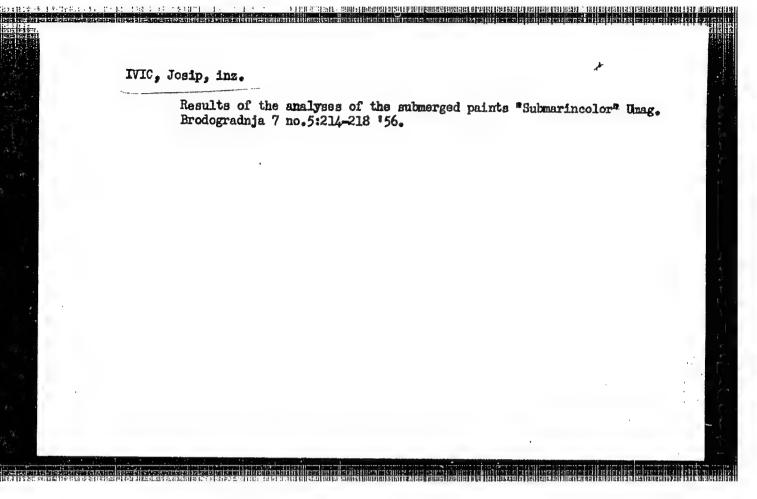
Bibliographical footnotes.

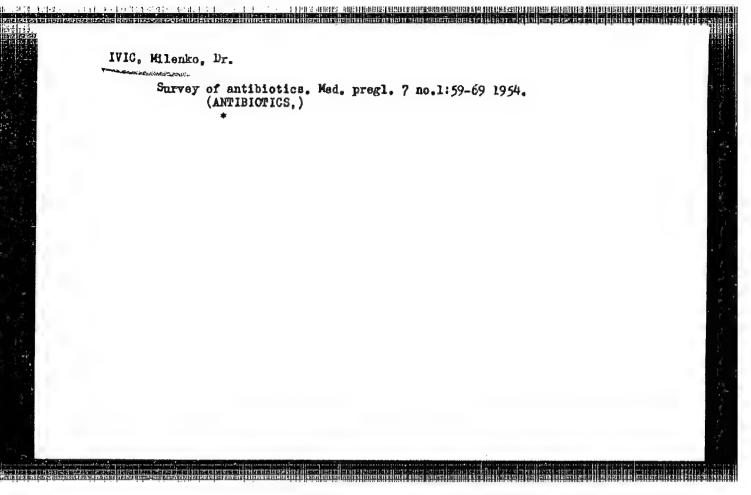
IVERIG, D.

Iron-sheet pipes. p. 1374
(Tehnika, Vol. 11, no. 9, 1956. Rockrad, Yugoslavia)

SG: Monthly List of East European Accessions. (MEAL) IC, Vol. 6, No. 7, July 1957. Uncl.







PUTNIK, Milan Dr; IVIC, Milanko. Dr.

Role of bacterial flora in strangulation of ileus. Med.pregl. Novi Sad 7 no.6:456-461 1954.

1. Pokrajinska medicinsko-istrasivacka laboratorija, Novi Sad.
Upravnik: Dr. Petar Svarc.
(INTESTINAL OBSTRUCTION, experimental,
bacteriol.aspects)

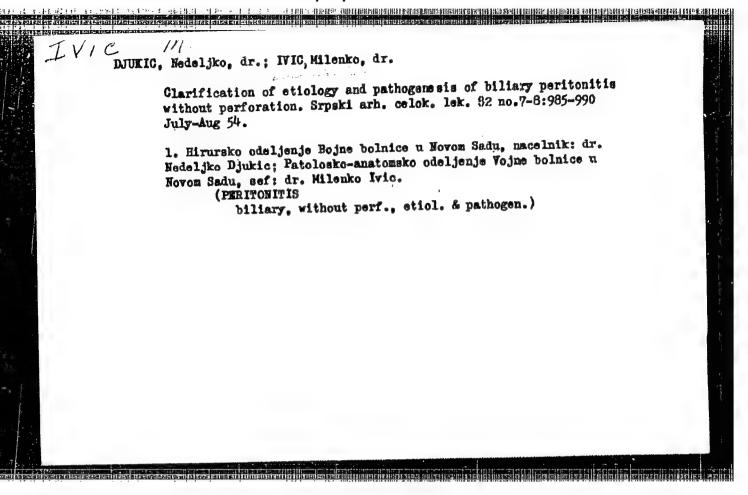
GASPAROV, Antun; IVIC, Milenko

Protein fractions in the serum and exudate in exudative pleurisy.
Srp arhiv lekar 82 no.2:181-189 F '54. (EEAL 3:7)

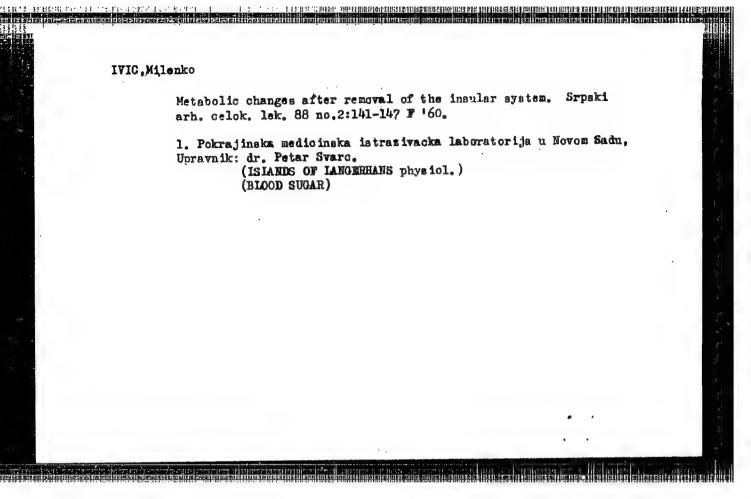
1. Interno odeljenje Vojne bolnice u Wovom Sadu, necelnik dr.
Antun Gosparov Klinicka laboratorija Vojne bolnice u Movom Sadu,
necelnik dr. Strahinja Marinkov. (Rad Je Urednistvo primilo 2-X1953 god.)

(BLOOD FROTBINS, in various dis.
\*pleurisy, exudative)
(KUUDATES AND TRANSUDATES
\*protein fractions in exudative pleurisy)
(PLEURIST

\*exudative, protein fractions in blood & exudate)



IVIC, H.  Insulinocytes A and diabetes. Acta med. iugosl. 13 no.4:433-441 159.	
1. Pokrajinska medicinska istrazivacka laboratorija u Novom Sadu.  (DIABETES MELLITUS pathol.)  (ISIANDS OF IANGERHANS pathol.)	
	5- 6-16-



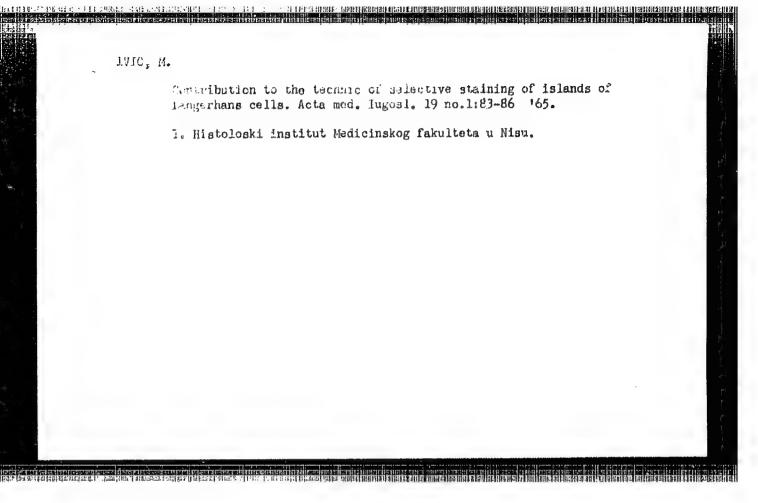
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On the problem of the site of synthesis of glucagon. Srpski arh. celok. lek. 89 no.10:1149-1153 0 161.

l. Institut za medicinska istrazivanja u Novom Sadu  $V_{\sigma}d$ . direktora: dr Petar Svarc.

(GLUCAGON metab)

E.



IVIC, Mijesira, visi grad. tehn. (Sarajevo, JNA 3)

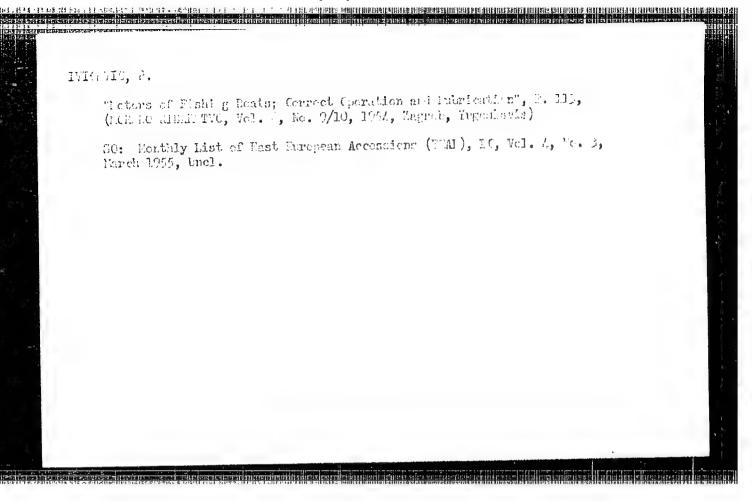
Construction of residential houses in Sarajevo. Tehnika Jug
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1. Direktor Fonda za stambenu izgradnju grada Sarajeva.

IVIC, Stepjan, inz., visi predavac

Profiles of the cadres in chemical industries. Glasnik hemicara BiH ll:85-96 162.

1. Telmoloski fakultet u Tuzli.



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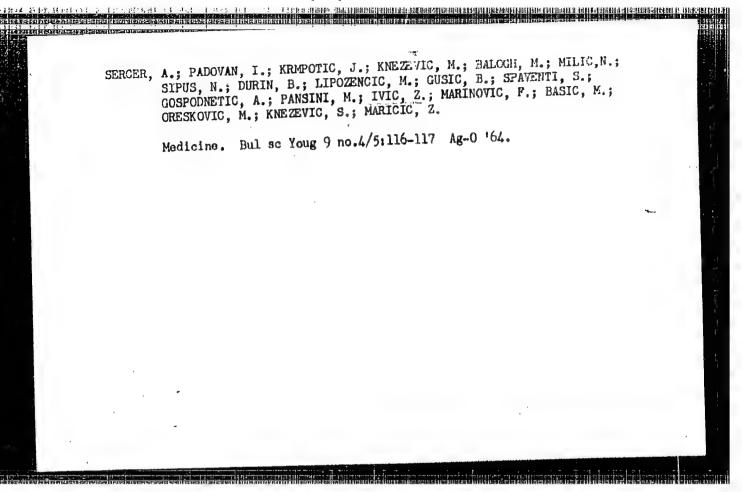
November 1955, Uncl.
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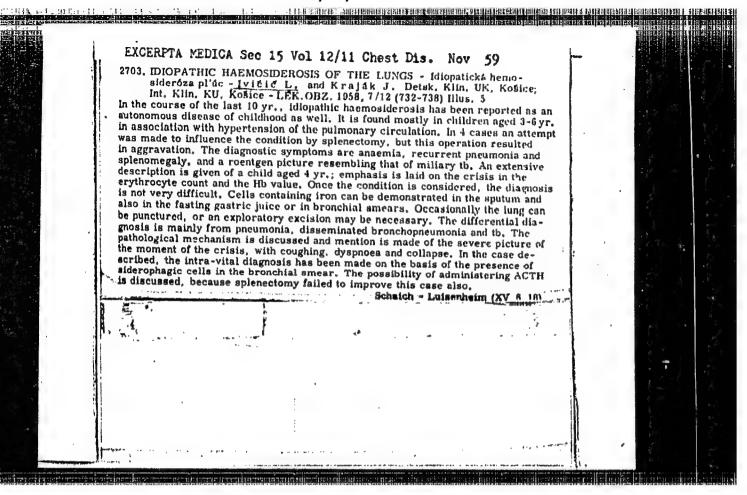
IVICIC, Ladislav, MUDr

Richtmograficersm of preumonia in newborn and older

Blectrocardiogram of pneumonia in newborn and older infants. Pediat. listy, Praha 9 no.5:265-266 Sept-Oct 54.

1. Z detskej kliniky IFSU v Kosiciach, prednosta Doc. MUDr. F.Demant (PNEUMONIA, in infant and child ECG)
(ELECTROCARDICGRAPHY, in various diseases pneumonia in inf. & child.)





#### CIA-RDP86-00513R000619320004-6 "APPROVED FOR RELEASE: 03/20/2001

HIMGARY / Chemical Technology. Chemical Products and H-5 Their Application -- Water Treatment. Sewage

Abs Jour: Ref Zhur-Khimiya, No 3, 1959, 8714

Author : Bartha, I., Ivicsics, F.

: Not given Inst

Title : Purification of Drainage Waters from the Petro-

leum Refining Industry

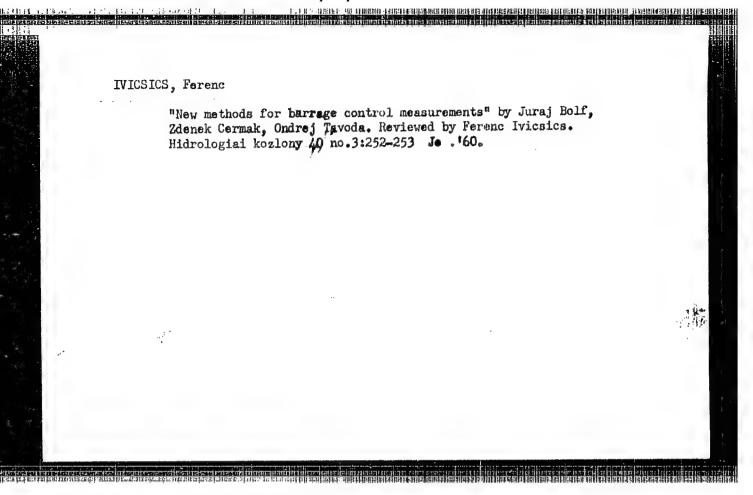
Orig Pub: Hidrol. kozlony, 1958, 38, No 1, 68-77

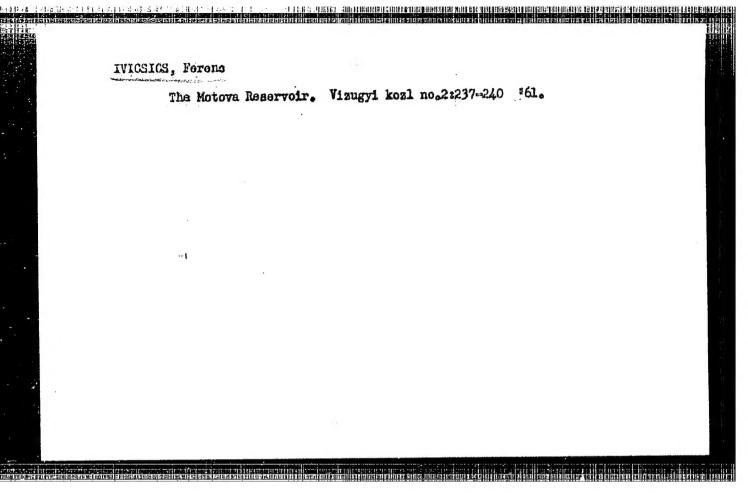
Abstract: A review of methods for computing petroleum traps

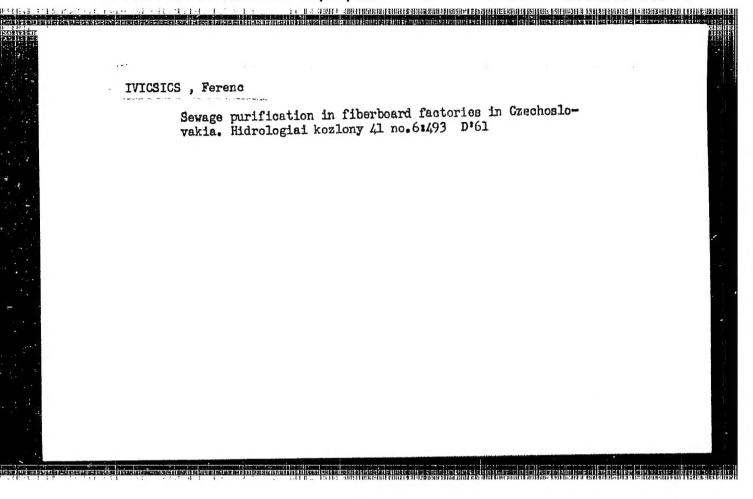
in accordance with Soviet and American practices.

Momograms are included. -- M. Lapshin

Card 1/1







IVICSICS, Ferenc, okl.mernok

Construction of dams in Czechoslovaki during the past 15 years. Vizugyi kozl no.1:158-162 '62.

1. A Vizgazdalkodasi Tudomanyos Kutato Intezet tudomanyos munkatarsa.

#### 

IVICSICS, Lajos, dr., okleveles mernok, a musseki tudomanyok kandidatusa, tudomanyos fomunkatars; IVICSICS, Fereno, okleveles mernok, tudomanyos munkatars

Solving similarity problems of hydraulic comptructions in connection with their structural model analysis. Vizugyi kozl no.2:268.276 162.

1. Scientific Research Institute of Water Resources Development, Budapest.